This article was downloaded by: On: 19 January 2011 Access details: Access Details: Free Access Publisher Taylor & Francis Informa Ltd Registered in England and Wales Registered Number: 1072954 Registered office: Mortimer House, 37- 41 Mortimer Street, London W1T 3JH, UK



### International Journal of Polymeric Materials

Publication details, including instructions for authors and subscription information: <http://www.informaworld.com/smpp/title~content=t713647664>

## Rheological Behavior of Polyethersulfone (PES)

Kun Qlª; Rui Huang<sup>ь</sup>

a Department of Materials Science and Engineering, Guangdong Institute of Technology, Guangzhou, P.R., China **b** Department of Plastics Engineering, Chengdu University of Science and Technology, Chengdu, Sichuan, P.R., China

To cite this Article Ql, Kun and Huang, Rui(1995) 'Rheological Behavior of Polyethersulfone (PES)', International Journal of Polymeric Materials, 29: 3, 157 — 164 To link to this Article: DOI: 10.1080/00914039508012109

URL: <http://dx.doi.org/10.1080/00914039508012109>

## PLEASE SCROLL DOWN FOR ARTICLE

Full terms and conditions of use:<http://www.informaworld.com/terms-and-conditions-of-access.pdf>

This article may be used for research, teaching and private study purposes. Any substantial or systematic reproduction, re-distribution, re-selling, loan or sub-licensing, systematic supply or distribution in any form to anyone is expressly forbidden.<br>distribution in any form to anyone is expressly forbidden.

The publisher does not give any warranty express or implied or make any representation that the contents will be complete or accurate or up to date. The accuracy of any instructions, formulae and drug doses should be independently verified with primary sources. The publisher shall not be liable for any loss, actions, claims, proceedings, demand or costs or damages whatsoever or howsoever caused arising directly or indirectly in connection with or arising out of the use of this material.

*Intern. J. Polymeric Mater..* **1995.** Vol. **29, pp. 157-164 Reprints available directly from the publisher Photocopying permitted by license only** 

*0* **1995 OPA (Overseas Publishers Association) Amsterdam** B.V. **Published under license by Gordon and Breach Science Publishers SA Printed in Malaysia** 

# Rheological Behavior of Polyethersulfone (PES)

*KUN QI\** 

*Department of Materials Science and Engineering, Guangdong Institute of Technology, Guangzhou* **570090,** *P.R. China* 

**and** 

#### *RUI HUANG*

*Department of Plastics Engineering, Chengdu University of Science and Technology, Chengdu 610065, Sichuan, P. R. China* 

*(Received December 3. 1994)* 

The rheological behavior of Polyethersulfone **(PES),** was studied using a Shimadzu Universal Materials Testing Machine and an Haake Torque Rheometer. It is found that in an industrially relevant range of shear rate and temperature **PES** was highly non-Newtonian. Its shear dependence and temperature dependence of viscosity were unexpected, and the power law index and activation energy were determined. **A** phenomenon **of** thickening of melt during processing **of PES** was found.

**KEY** WORDS Polyethersulfone (PES), rheological property, Haake Plasticorder, thickening of melt.

#### **1. INTRODUCTION**

Polyethersulfone (PES) is perhaps one of the most important engineering thermoplastics that displays excellent properties.<sup>1.2</sup> The use of this material requires that it be melt processed for injection and extrusion molding, etc. In previous papers there are a lot of brief introduction about the processing conditions of **PES3,"**  and seldom deals with the rheological properties.' But no study has been conducted to investigate the rheological properties of **PES** relevant to real processing conditions. The knowledge of the rheological behavior of PES over an industrially relevant range of shear rate and temperature is essential for the assessment **of** the material processability , process design optimation and troubleshooting. It is to this end that this work was undertaken.

The rheological behavior of PES over an industrially relevant range of shear rate

<sup>\*</sup>To whom correspondence should be addressed.

and temperature has studied by means of a capillary instrument attached to a Shimadzu Universal Material Testing Machine. **As** a torque rheometer (plasticorder), this rheological device is useful for studying the rheological behavior of plastics. It produces a flow curve that is a measure **of** the resin's processibility. **A torque-ternperature-time** curve of PES, measured by Haake microcomputer-controlled torque rheometer model *SYS* 40, is presented in this paper and can be directly related to the real world of molding.

#### **2. EXPERIMENTAL**

#### **2.1 Materials**

The PES resin used in this study was kindly provided by Jilin University, Changchun, **P.R.** China. Its molecular Weights is *Mn* = 30000 (data supplied by supplier **of**  PES resin). Prior to use in experiments, **PES** pellets (diameter *s 3* mm) were dried in a vacuum furnace at 150°C for 4 h to remove any absorbed moisture.

#### **2.2 Tests and Measurements**

Melt rheological properties of **PES** were evaluated on a capillary instrument attached to a Shimadzu Universal Materials Testing Machine model **AG-IOTA. A**  capillary die of 1 mm in diameter and **10** mm in length was used. The applied pressure forces, *F,* were recorded on the Shimadzu at loading speeds, *V,* of *5,* 10, **20,** 50, 80, 100, 150, 200, 300 and 500 mmlmin and at temperatures 315, 330 and 350°C.

Kneading of **PES** performed on an Haake Torque Rheometer model *SYS* 40, fitted with a RHEOMIX 600 mixer, at a set temperature of 335°C and at a rotor speed of 15, 25 and 40 rpm for 1 h. The torque required to turn the Haake and the material temperature were recorded and processed automatically by a microcomputer attached to the Haake as a function of time.

#### **2.3 Rheological Analysis**

The capillary rheological data were processed as follow:

The apparent shear rate,  $\gamma_w$ , were expressed by

$$
\gamma_w' = 4Q/\pi R^3
$$

where *R* is the capillary die radius  $(=0.5 \text{ mm})$  and *Q* is the volumetric flow rate which can be expressed as

$$
Q = \pi R_p^2 V
$$

where  $R_p$  is barrel radius (= 5 mm) and *V* is the loading Speed (mm/min).

Thus, the apparent shear rate could be calculated from

$$
\gamma_w' = 4\pi R_p^2 V/\pi R^3 = 40V/3 \text{ (l/s)}
$$

The shear stress at the wall of capillary were calculated **by** 

$$
\tau_w = R\Delta P/2L = 318.3F \text{ (Pa)}
$$

where *F* is the applied pressure force  $(N)$ , *L* is the capillary die length (= 10 mm) and  $\Delta P$  is the pressure differential on the end of capillary die. So, the apparent viscosity,  $\eta_a$ , were calculated with the value of the apparent shear rate,  $\gamma_w$ , and shear stress,  $\tau_w$ , as follow

$$
\eta_a = \tau_w / \gamma_w' = 23.873 F / V \text{ (Pa·s)}
$$

and the equation used for calculation of the power law index, *n,* is expressed as

$$
n = \delta \log \tau_w / \delta \log \gamma_w'
$$

The Bagley and Rabinowitsch corrections were not applied to the capillary rheometry data.

#### **3. RESULTS AND DISCUSSION**

#### **3.1 Shear Rheometry of PES Melt**

*3.1.1* Flow *curves.* Viscosity curves of PES measured at 315, 330 and 350°C and for shear rates from 10 to 10000 *11s* are presented in Figure **1. A** typical pseudoplastic behavior can be seen. That is the melt viscosites of PES decrease with the increase of apparent shear rates.

The shear stress vs. rate plots of molten PES are shown in Figure 2. **As** expected, due to the strong non-Newtonian behavior, the variations of log (shear rate) with log (shear stress) are not linear (Figure 2). It suggests that the power law index, *n,* will change with the shear rate. The power law index values calculated are presented in Table **I.** 

The power law indices, *n*, in Table I are close to the value  $(n = 0.53)$  reported by Saini.' **As** can be seen, the values of *n* decrease with decrease in temperature and increase in shear rate. In other words, the lower the temperature and the higher the shear rates were, the more non-Newtonian behavior of the molten PES would be. It is well consistency with Saini.<sup>5</sup>

*3.1.2 Factors affecting rheological properties of* PES. The temperature dependence **of** the viscosity was expressed in terms of an Arrhenius equation

 $q_a = A \exp(E/RT)$  or  $\log q_a = \log A + (E/2.303R) * (1/T)$ 



FIGURE 2 Shear stress vs. shear rate for PES.

The flow activation energy (J/mol),  $E$ , for viscous flow can be measured from log (apparent viscosity) vs.  $1/T$  plots, where T is the absolute temperature  $(K)$  and R is a constant  $(=8.314 \text{ J/K}\cdot \text{mol})$ . The activation energy of viscous flow is found to decrease with increase in shear rate (Figure 3). But it can be seen that viscosity of molten **PES** has a weak sensitivity to temperature, because of its fairly low viscous **flow** activation energy.

As seen above, the viscosity of **PES** decreases with increase in shear rate. **PES**  consists of semi-rigid molecular chains which has strong effect on its rheological behavior. With a semi-rigid-chain PES, the decrease in apparent viscosity with



Temperature (C)	Flow index values at shear rates of	
	$10 - 999$ 1/s	$1000 - 10000$ $1/s$
315	0.50	0.33
330	0.58	0.41
350	0.63	0.40

**Flow index values of PES** 



**FIGURE 3 Variation of viscous flow activation energies with shear rate** 

increasing shear rate, is less than that of flexible-chain polymer, but higher than that of a rigid-chain polymer.

3.1.3 Effect of *rheological behavior on molding*. In plastics processing, adequate melt viscosities are required to make products using different processing methods. By adjusting the molding conditions, the processing flexibility can be enhanced.

Because of its molecular chain structure, the viscosity of **PES** is not very sensitive to changes in temperature or shear rate. **So** the flow performance cannot be increased effectively by simple increases in shear rate (screw rpm) or increases in barrel temperature.

#### **3.2 Torque Rheological Property of PES Melt**

The rheological characterization of the **PES** resin was made by using capillary and torque rheometer. The torque-temperature-time diagram of **PES** at a set temperature of **335°C** are presented in Figure 4. The results **of** kneading test are summarized in Table 11.

Torque rheometer produces a flow curve that **is** a measure of the polymer's pro~essability.~.' In Figure **4,** the curve **of** TQ and **T2** represent the relationship between torque and temperature with time. The point of L, **S** and D were the times of loading, stabilization and degradation of **PES,** recorded by Haake, during kneading. It can **be** seen from Figure 4, that the torque upon loading of **PES** is quite large, but stabilized to a relative small value. *So,* the energy consumption of molding machine during processing of **PES** would not be very large. This has been proved by our extrusion and injection molding tests.<sup>7,8</sup> The time of relative high torque (about point **L)** is very short.

At the present condition, the torque and temperature stabilize to a constant value in about **5** min (Figure **4).** This suggests that in order to obtain a well molten **PES,** the molding machine with a high length to diameter ratio, *LID,* twin-screw and/or high pressure should be used.

At time (point D) of over **40** min, a considerable increase of torque is observed. The color of **PES** products turns black and energy consumption of machine arises



**FIGURE 4** Record **torque (in** meter-gram) **vs. the** time at **a** set temperature *335°C* and at 15 rpm.

**TABLE I1**  Results of kneading **tests** 



**Unit conversion: 1mg=9.807 X 10-"N** 



**FIGURE** 5 **Variation of torque with rpm at 335°C.** 

during injection and extrusion molding, if the melt has a long residence time in the barrel.<sup>8</sup> The increase in melt viscosity with the increase in processing time, caused by prolonged shear, heat and oxidation is different from the thermal degradation. This phenomenon had been named "thickening of melt during processing of **PES"** by present authors.'

There is a relationship' between torque *(Nrn), M,* and rotor speed (rpm), *S,* 

$$
M = C S^{\alpha}
$$

where C and  $\alpha$  are constants. This relationship resembles the familiar power law and can be used for calculation of the power index, *n.* Figure *5* shows variation of equilibrium torque (torque at point **S** as in Figure 4) vs. rpm at *335°C.* A power index obtained from Figure 5 is  $n = 0.58$ , which is consistent with the data of Saini<sup>5</sup> and capillary rheological results presented above.

#### **4. CONCLUSIONS**

Capillary rheological experiments show that molten **PES,** as a typical pseudoplastic fluid, has a small non-Newtonian index  $(n = 0.33 - 0.63)$  and a rather low viscous flow activation energy  $(E = 30-70 \text{ kJ/mol})$ . And hence, the shear rate dependence and the temperature dependence of the viscosity are both not obvious.

According to the torque-temperature-time plots measured under industrial relevant processing condition, the energy consumption during molding would be acceptable. But the residence time of resin in the barrel must be less than 40 min. **A** thickening of melt during processing of **PES** takes place.

#### **References**

- **1.** *0.* **B. Searle and R. H. Pfeiffer,** *Poiym. Eng. Sci., 25,* 474 **(1985).**
- 2. **J. P. Critchley,** *G.* **J. Knight and W. W. Wright, "Heat-Resistant Polymers," Plenum Press, New York, 1983.**
- **3. G. Blinne, M. Knoll, D. Muller and K. Schlichting,** *Kunsfsfoffe,* **75, 29 (1985).**
- **4. E. Doring,** *Kunsfstoffe,* **80, 1149 (1990).**
- **5. D. R. Saini and A. V. Shenoy,** *J. Elastomers Plastics,* **17, 189 (1985).**
- **6. Donald V. Rosato and Dorninick V. Rosato, "Plastics Processing Data Handbook," Van Nostrand Reinhold, New York, 1990, Chap. 10, pp. 331-342.**
- **7. Kun Qi and R. Huang,** *Polym. Plast. Technol. Eng., 33,* **121-133 (1994).**
- **8. Kun Qi, S. Lai, Z. Zhang and R. Huang,** *China Plastics (Chinese), 6,* **34 (1992).**
- **9. I. Mathew, K.** E. **George and D. J. Francis,** *Intern. J. Polymeric. Muter.,* **21, 189 (1993).**